

Three ways to grow faster and better CIGSe:

**In-Ga co-electrodeposition,
1 second laser annealing,
and Cu-rich growth**

Phillip Dale¹, Joao Malaquias¹, Helen Meadows¹, Marc Steichen¹
Valérie Deprédurand², Yasuhiro Aida², Susanne Siebentritt²,
Ashish Bhatia³ and Mike Scarpulla³

(1) Laboratory for Energy Materials, University of Luxembourg

(2) Laboratory for Photovoltaics, University of Luxembourg

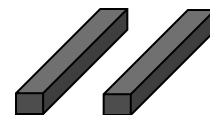
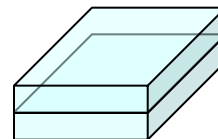
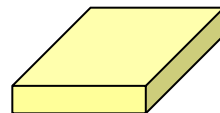
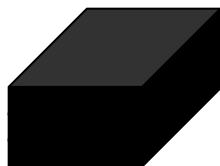
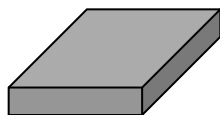
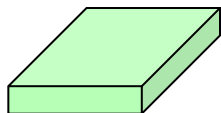
(3) Department of Materials Science and Engineering,
University of Utah, USA



Acknowledgements



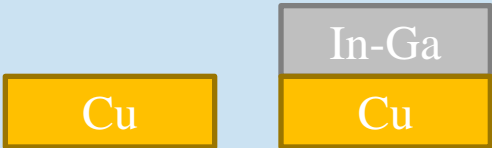
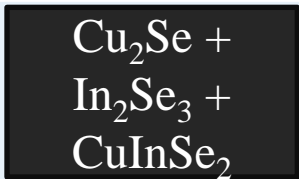

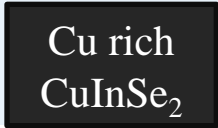

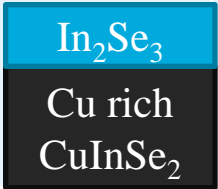
Luxembourg baseline process



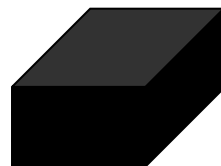
Soda Lime Glass	Mo	Cu(In,Ga)Se ₂ Cu ₂ ZnSnSe ₄	CdS	i-ZnO Al:ZnO	Ni/Al
	DC sputter	Co-evaporation / Electrodeposition and annealing	CBD	Rf – sputter	E-beam evaporation
25 x 25 x 2 mm	500 nm	1 – 3 μm	50 nm	80 nm 500 nm	10 nm 2000 nm

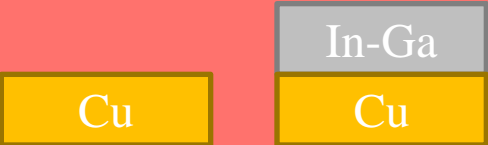
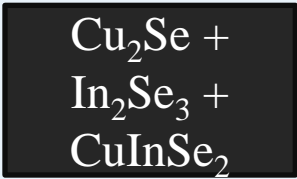

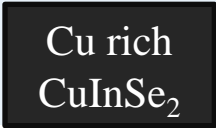

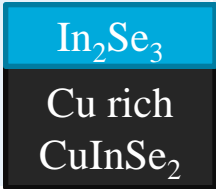
Luxembourg Cu(In,Ga)Se₂ processes



	Electrodeposition		Annealing	Advantage
1			Elemental selenium Tube furnace / RTP	Good control of Ga / III
2			 1064 nm Nd:Yag Laser	Very fast annealing
	Co-evaporation			
3			 	Less defects

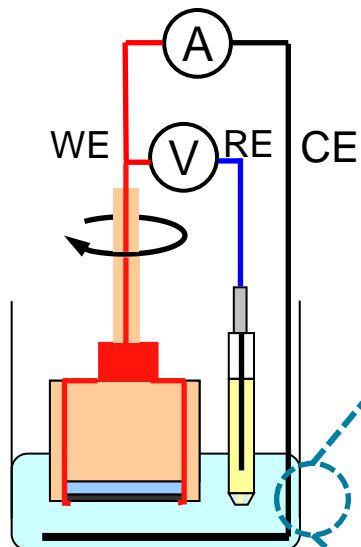
Talk outline



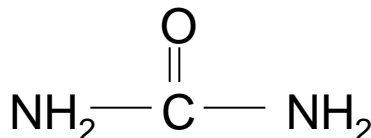
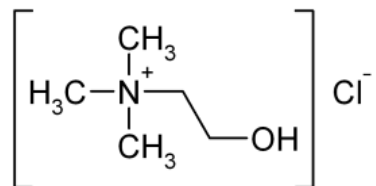
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Electrodeposition and annealing

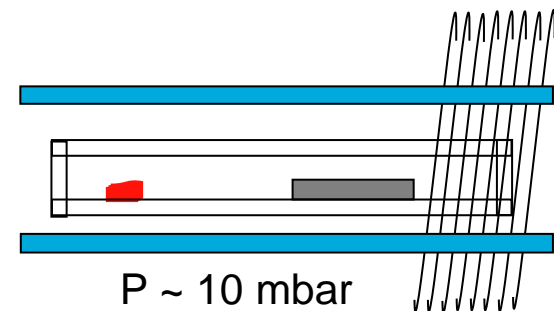
Electrodeposition of metals



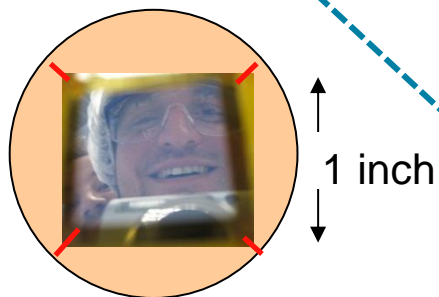
Solvent



Tube furnace
 $\Delta T \sim 33 \text{ }^\circ\text{C min}^{-1}$



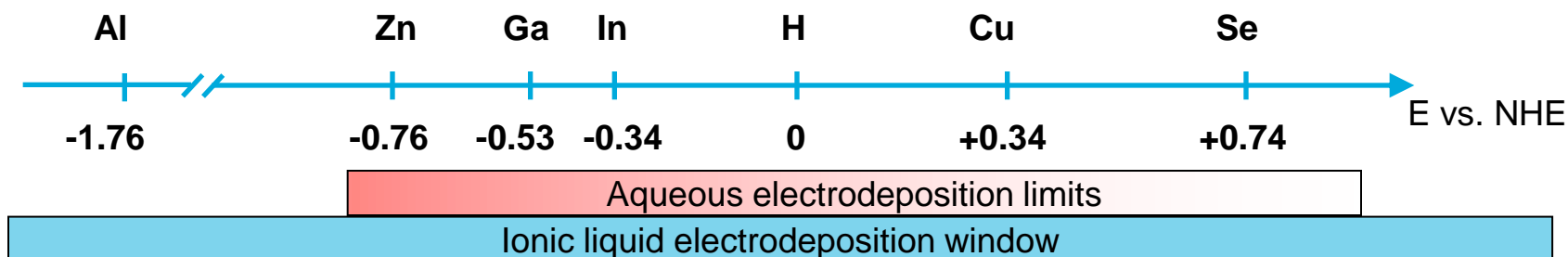
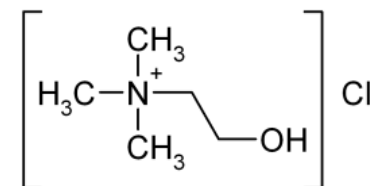
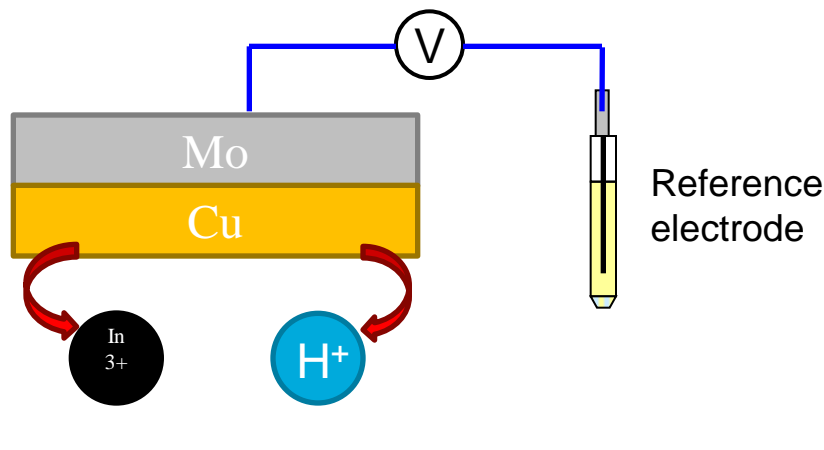
N_2, H_2



Mo/Cu/In:Ga

1. Co-electrodeposition of In and Ga could lead to more uniform electrodeposits on the microscale?
2. Could lead to faster deposition times.

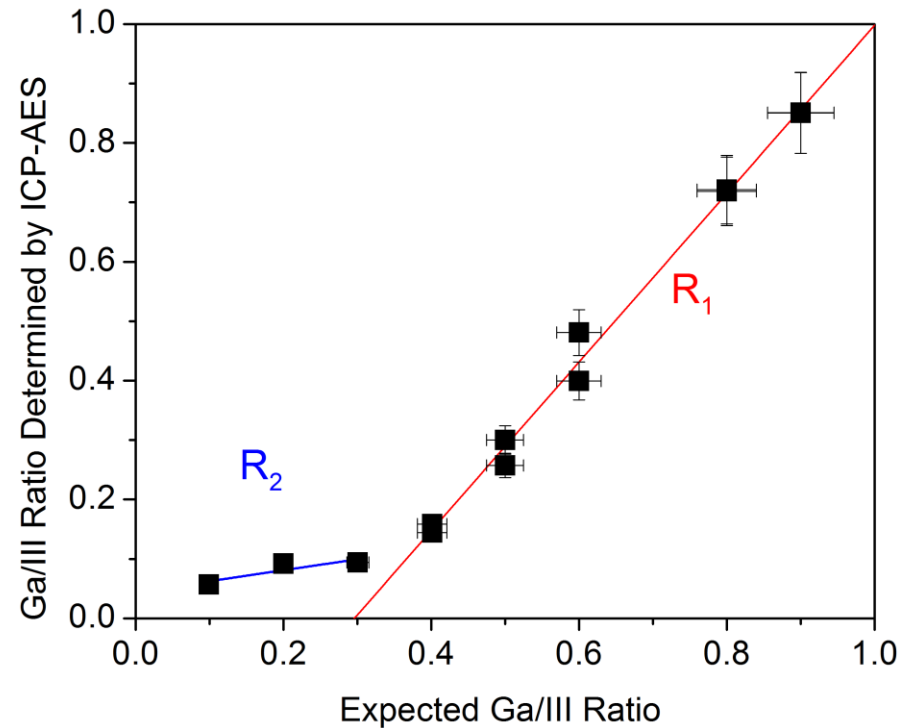
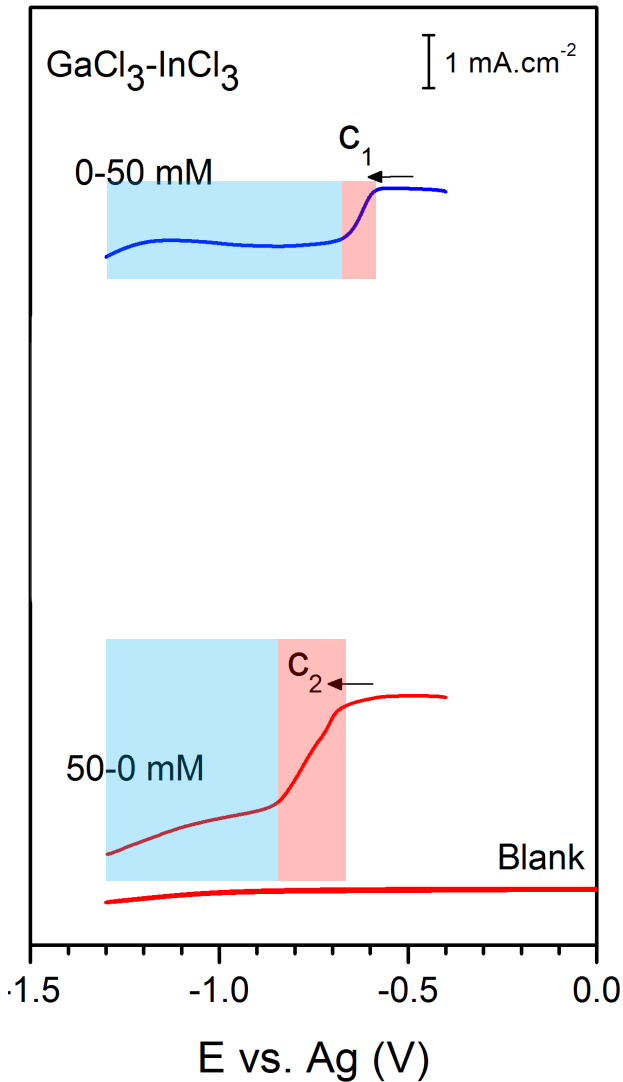
Advantages of electrodeposition from ionic liquids



Wider electrochemical window enables facile deposition of low reduction potential elements like Ga

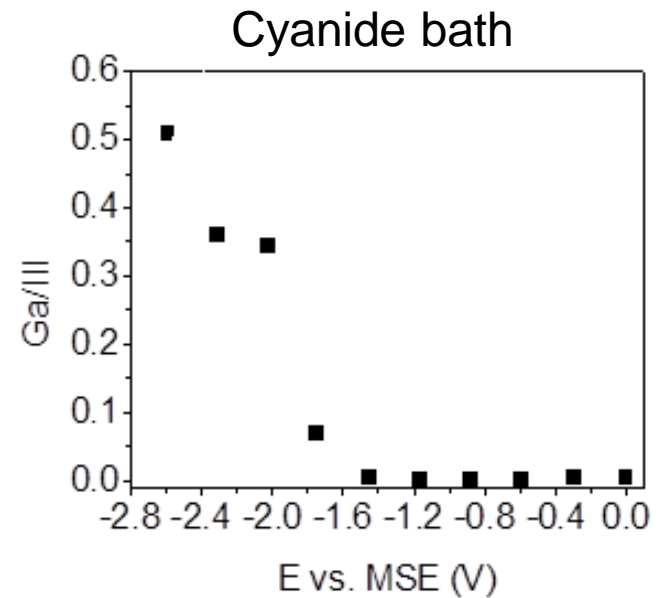
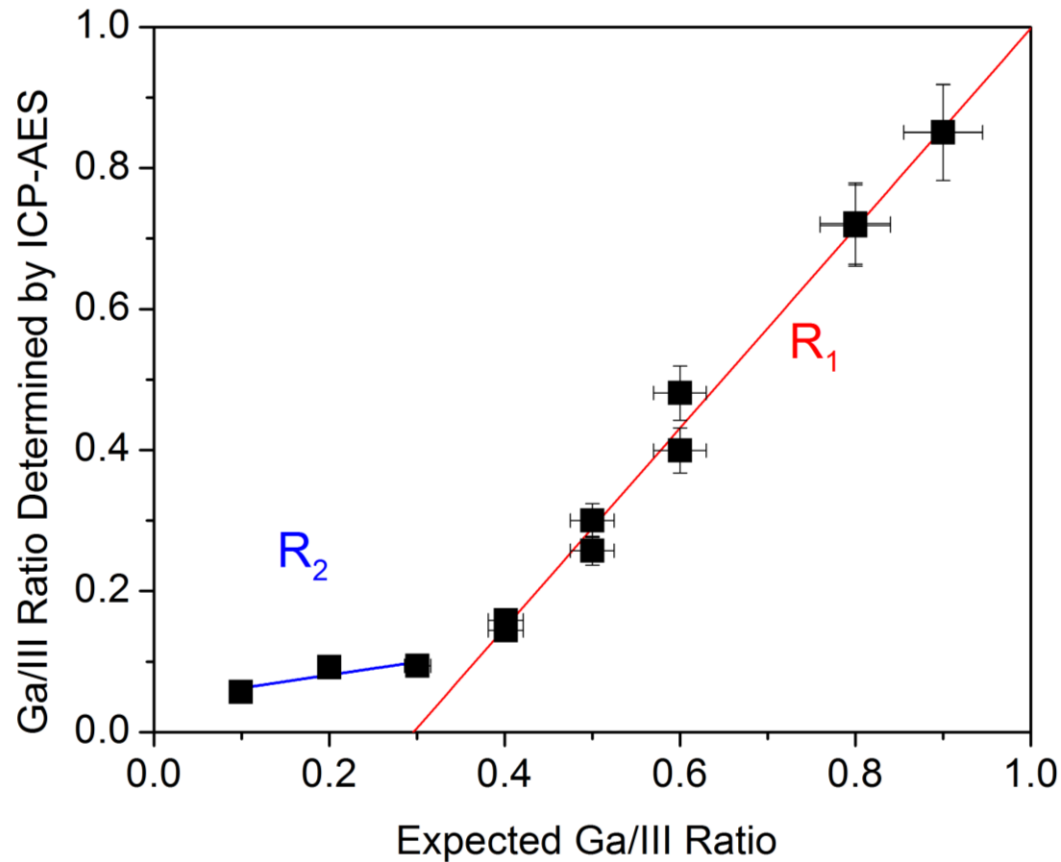
Steichen, Thomassey, Siebentritt, Dale . *Phys. Chem. Chem. Phys.* **13** 4292-302

Co-electrodeposition of In and Ga



Malaquias et al. Manuscript in preparation

Comparison to academic literature



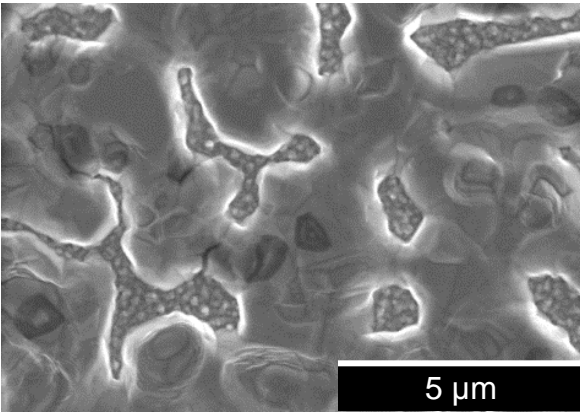
Adapted from Zank et al., *Thin Solid Films*, 1996 , **286**, 259

Coelectrodeposition of In-Ga on Cu

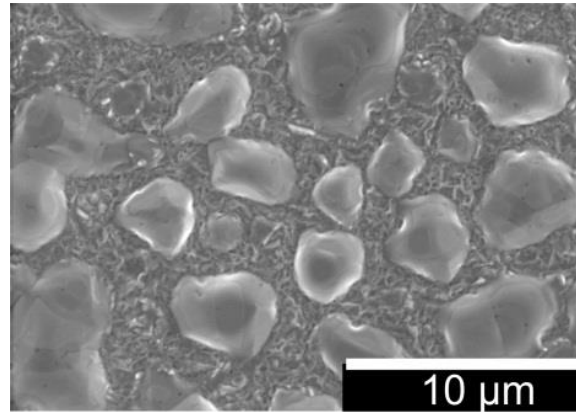
→ Depositions at $E_d = -1.2$ V ; constant Q; $Ga/III_{obtained} = 0.1; 0.4; 0.7$; $T = 60$ °C

- SEM/EDX

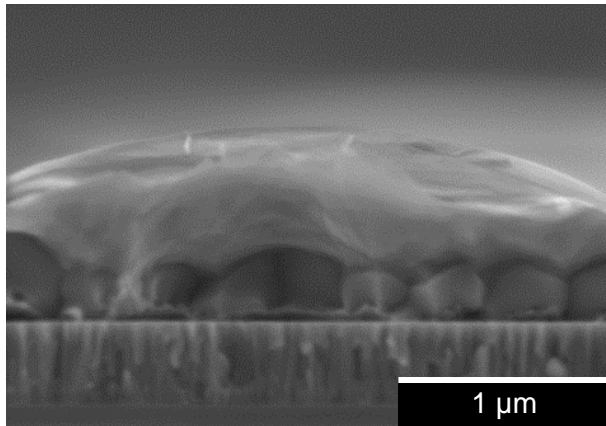
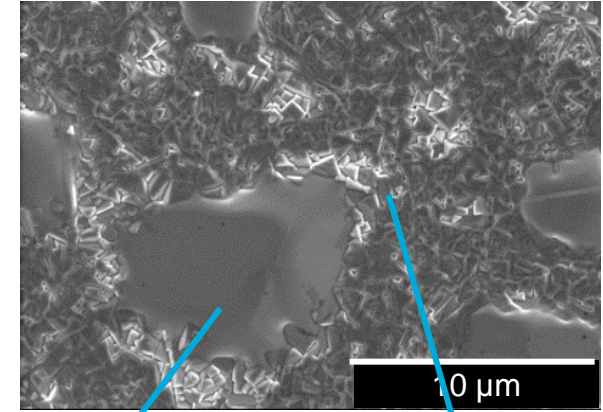
Ga/III = 0.1



Ga/III = 0.4



Ga/III = 0.7



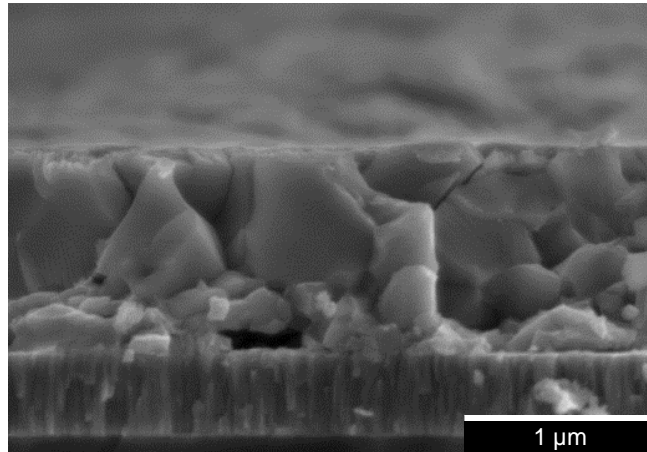
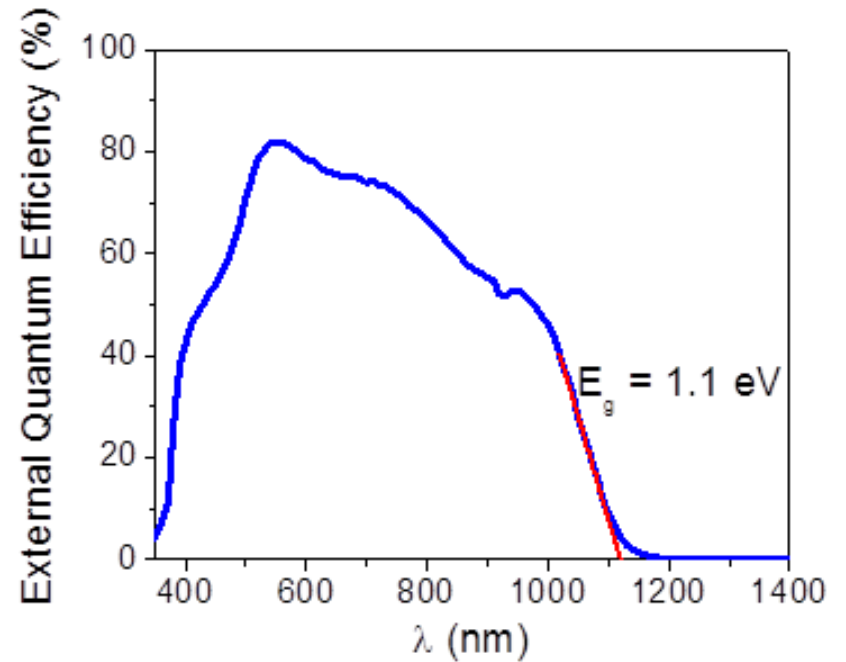
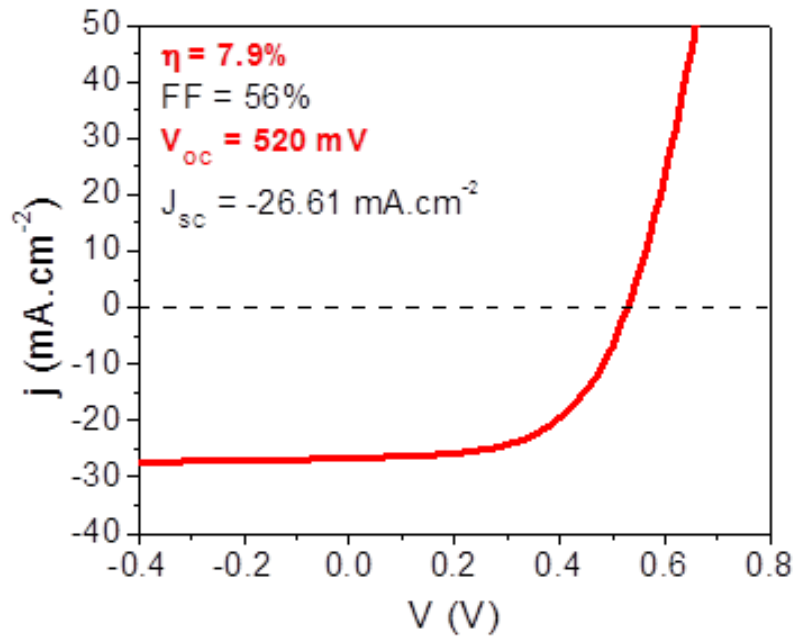
EDX:

Element	Atomic %
Cu	4.6
In	88.5
Ga	4.8

Element	Atomic %
Cu	42.6
In	0.5
Ga	26.5

Best Preliminary Device Results



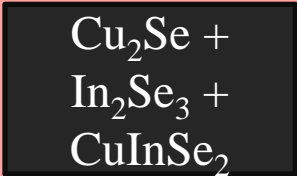
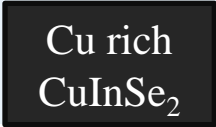
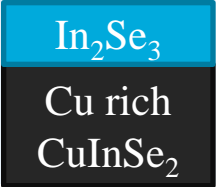
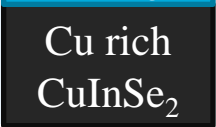
Ga/III = 0.4



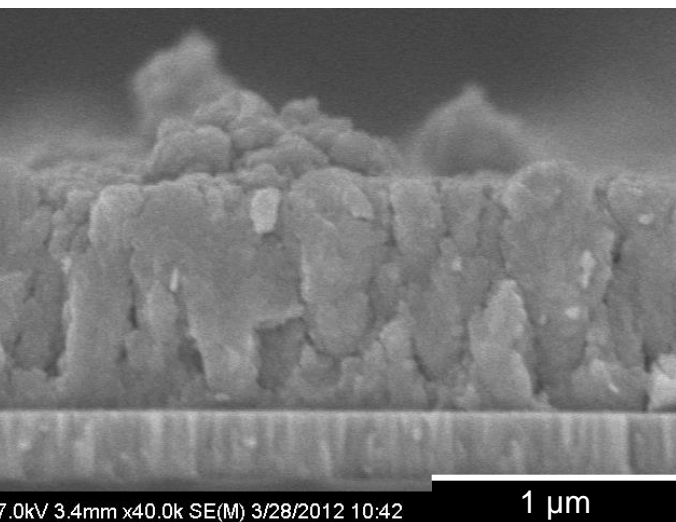
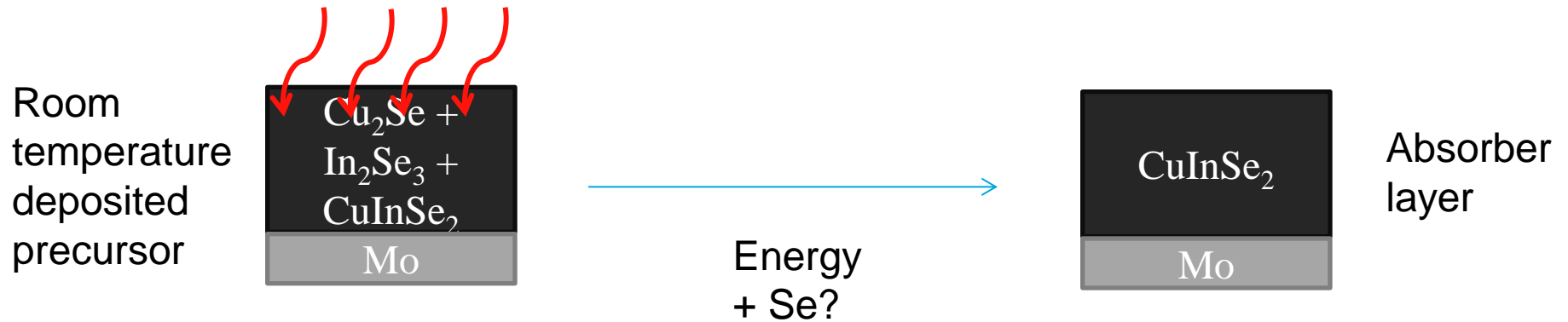
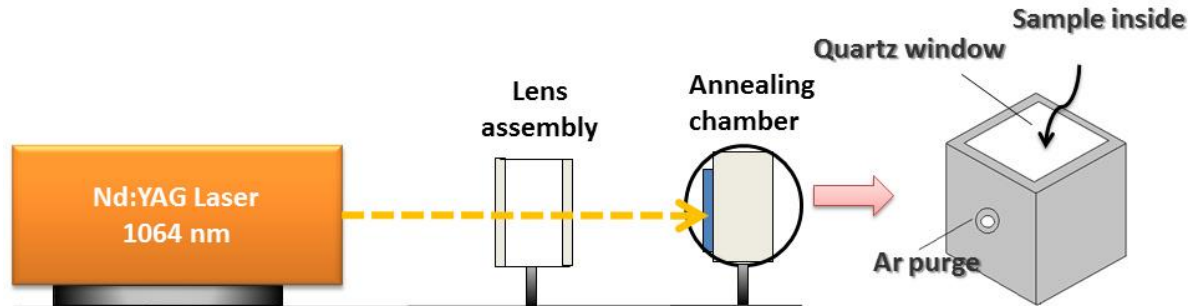
Malaquias et al. Manuscript in preparation

Talk outline



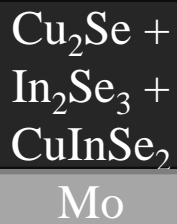
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3			 	Less defects

Electrodeposition and Laser Annealing



Advantages of Laser Annealing?





1. heat just the local environment
2. absorber layer temperature can exceed glass softening temperature on sub milli-second timescales



Is it possible to anneal an absorber in 1 s?

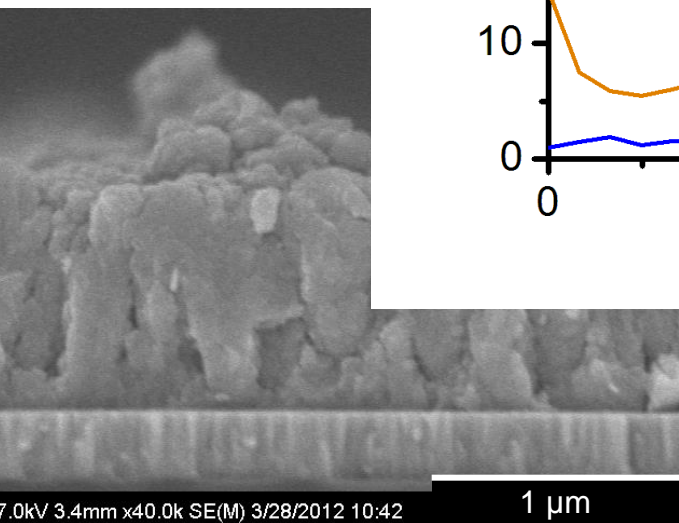
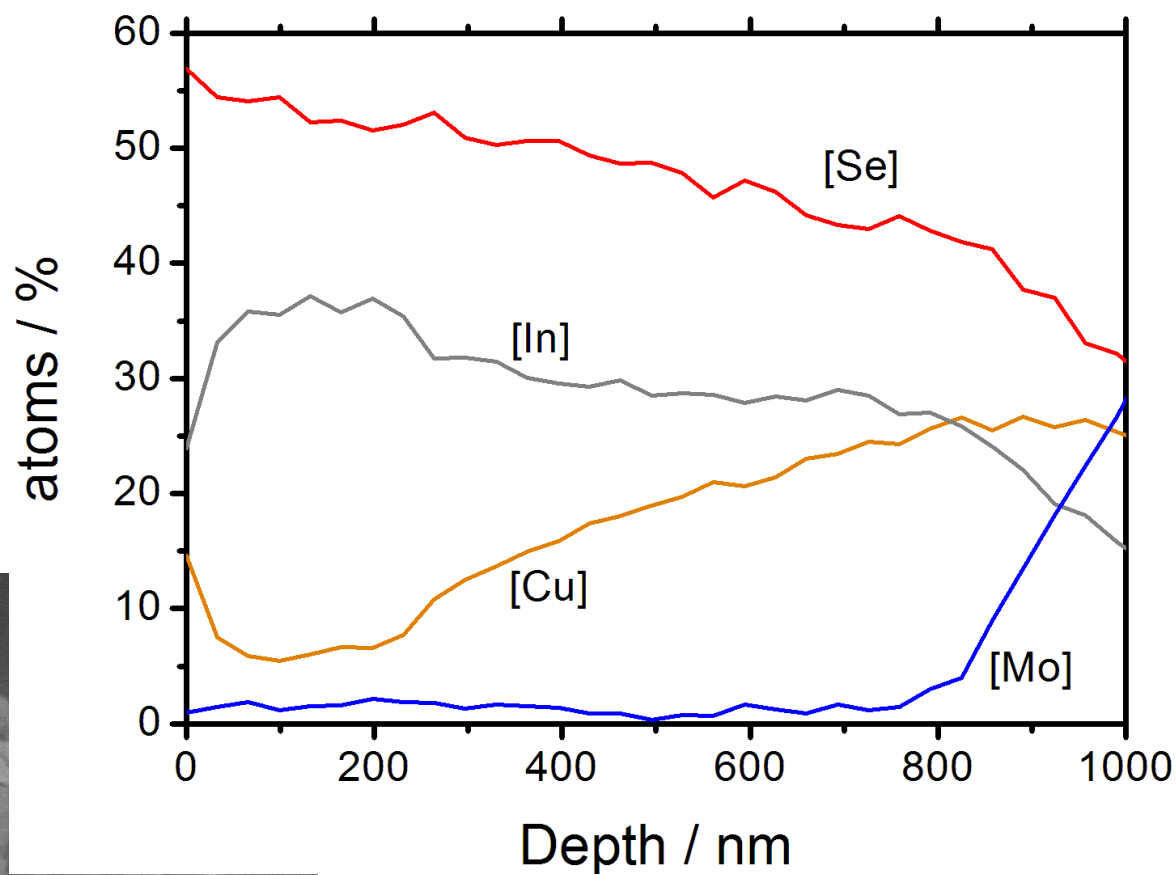
What does annealing do?

Energy side

1. diffuse atoms to remove concentration gradients (Cu, In, Se)  chemical depth profiles
 2. grow grains
 3. drive chemical reactions
 4. move defects to grain boundaries
 5. move Na or O to pacify defects at grain boundaries
-   x-ray diffraction
-  ? opto-electronic properties

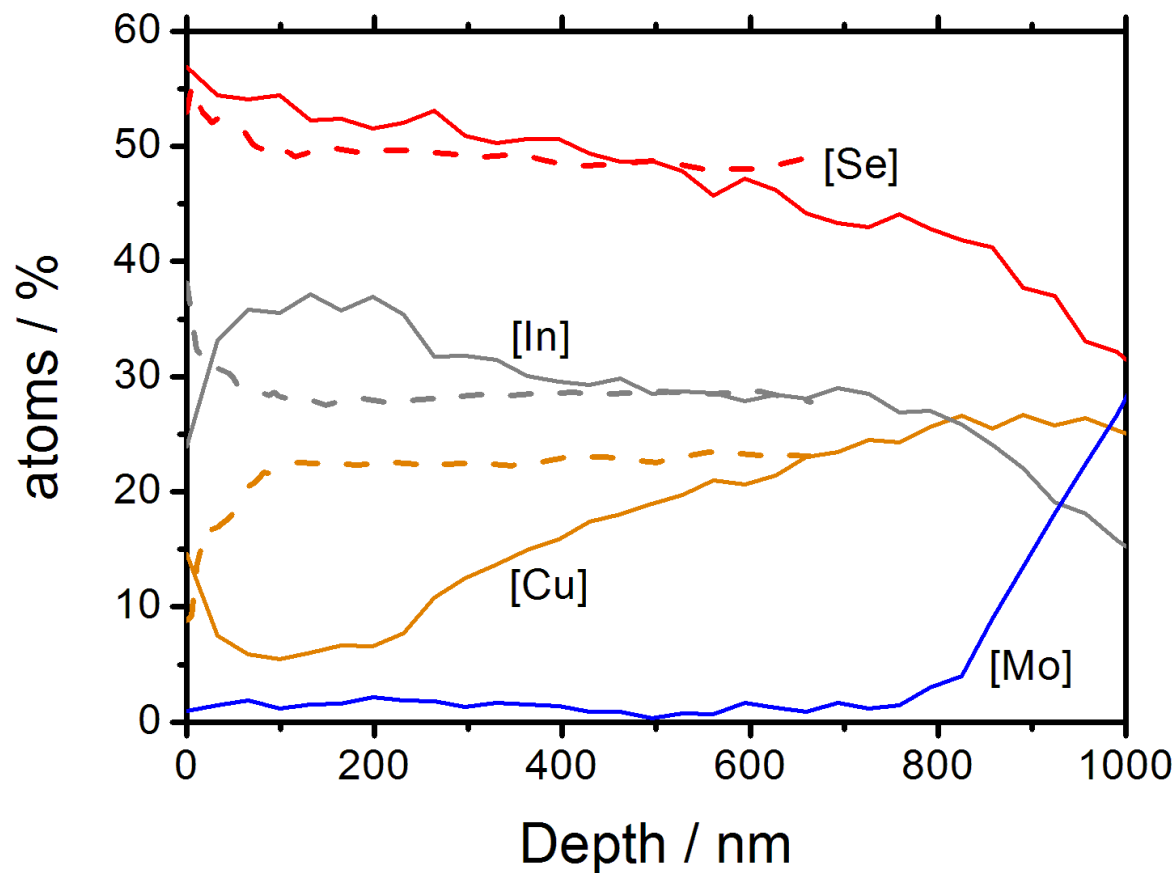
$\text{Cu}_2\text{Se} +$
 $\text{In}_2\text{Se}_3 +$
 CuInSe_2
Mo

Precursor



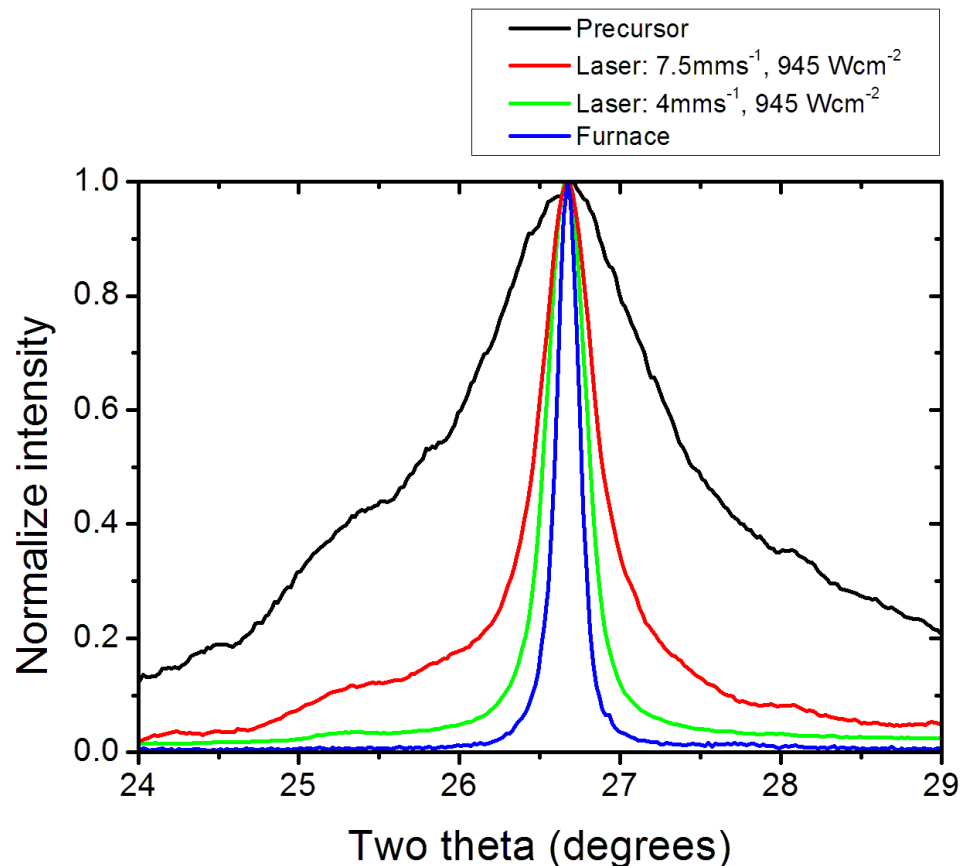
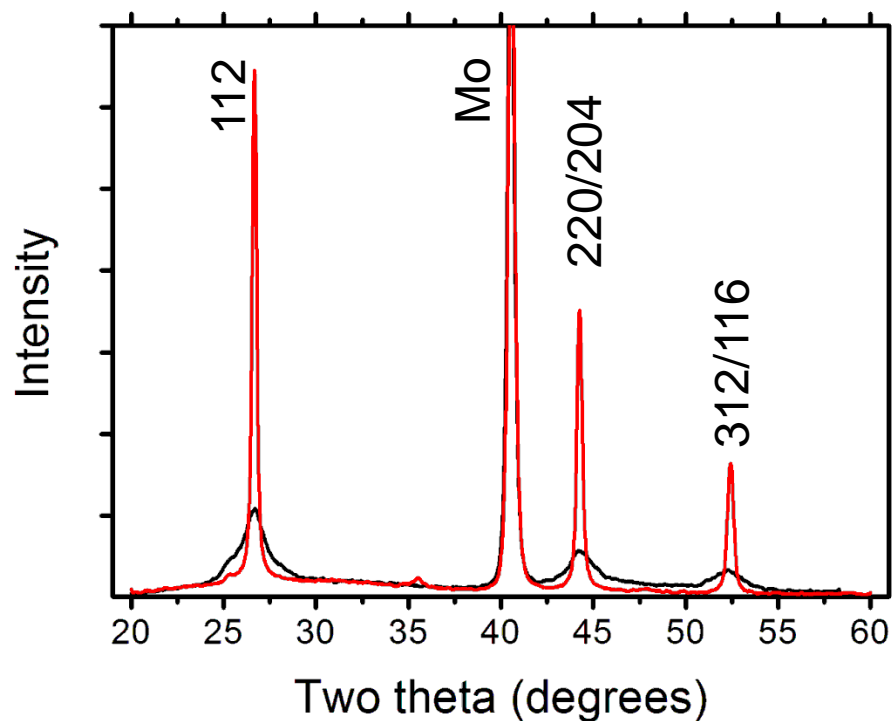
$\text{Cu}_2\text{Se} +$
 $\text{In}_2\text{Se}_3 +$
 CuInSe_2
Mo

Laser Annealing for 1s at 945 W.cm^{-2}



$\text{Cu}_2\text{Se} +$
 $\text{In}_2\text{Se}_3 +$
 CuInSe_2
Mo

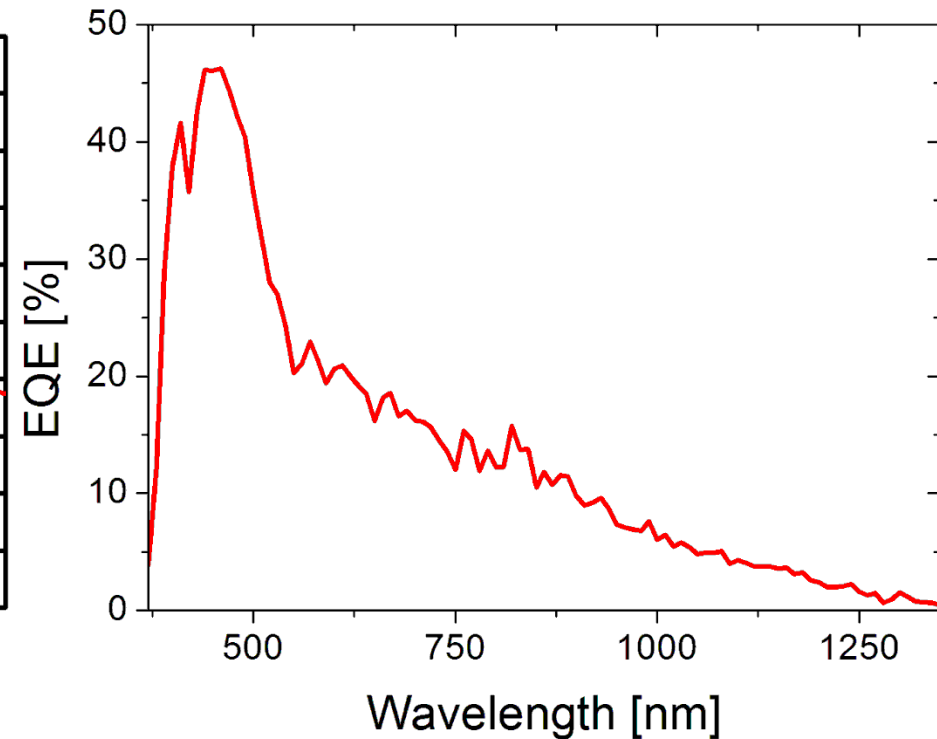
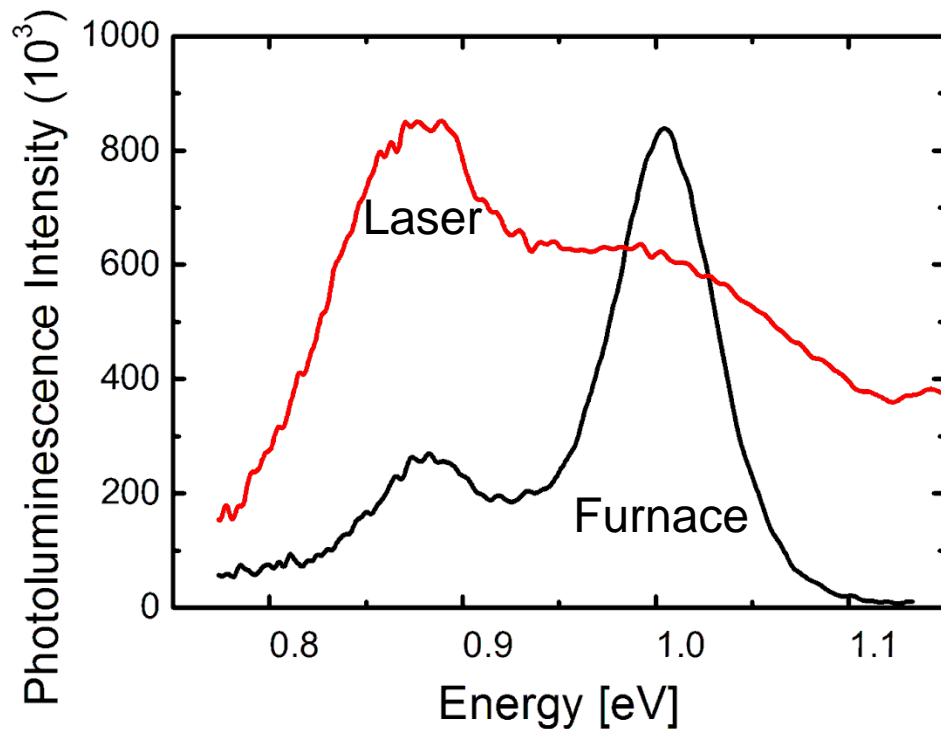
Laser Annealing at 945 W.cm^{-2}



FWHM of main peaks decreased after laser annealing
indicating increased crystal coherence length

$\text{Cu}_2\text{Se} +$
 $\text{In}_2\text{Se}_3 +$
 CuInSe_2
Mo

Opto – electronic properties

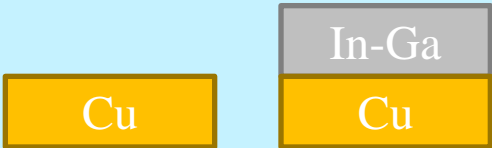
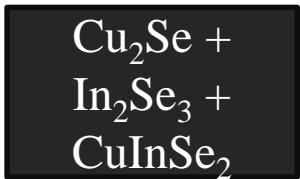

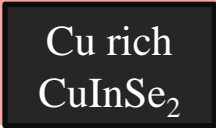

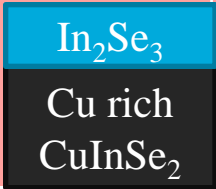


Meadows et al. manuscript in preparation

Laser annealed absorber layer shows PL properties
When completed to full device, shows rectification...

Talk outline



	Electrodeposition		Annealing	Advantage
1			Elemental selenium Tube furnace / RTP	Good control of Ga / III
2			 1064 nm Nd:Yag Laser	Very fast annealing
	Co-evaporation			
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Quasi fermi level splitting in absorber indicates maximal V_{oc} of device

PL

Device, large $E_{fn} - E_{fp}$

smaller $E_{fn} - E_{fp}$

E_c
 E_{fn1}
 E_{fn2}

E_{fp}
 E_v

p

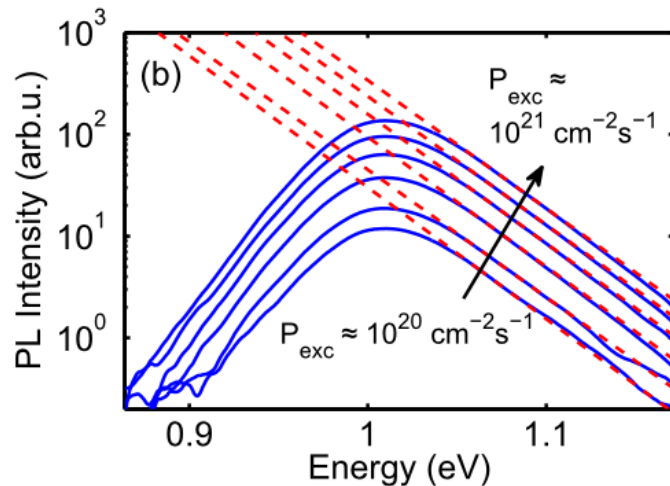
n

p

n

V_{oc}

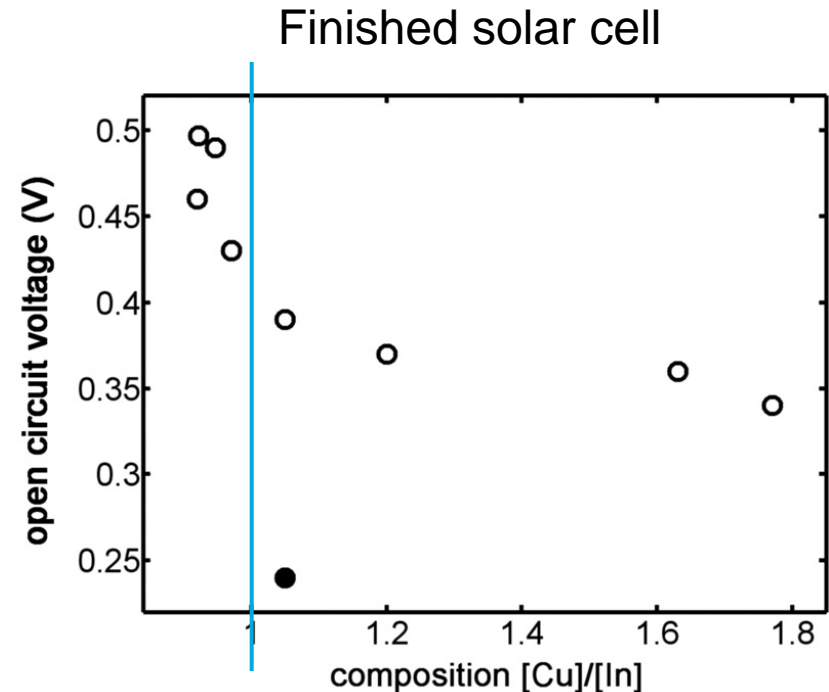
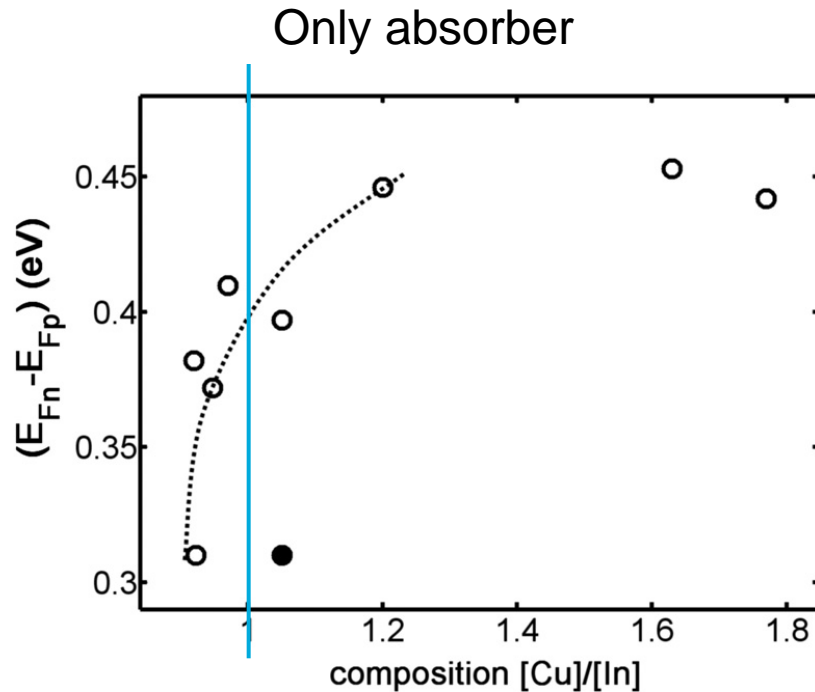
V_{oc}



higher V_{oc}

lower V_{oc}

Quasi-Fermi level splitting in polycrystalline layers

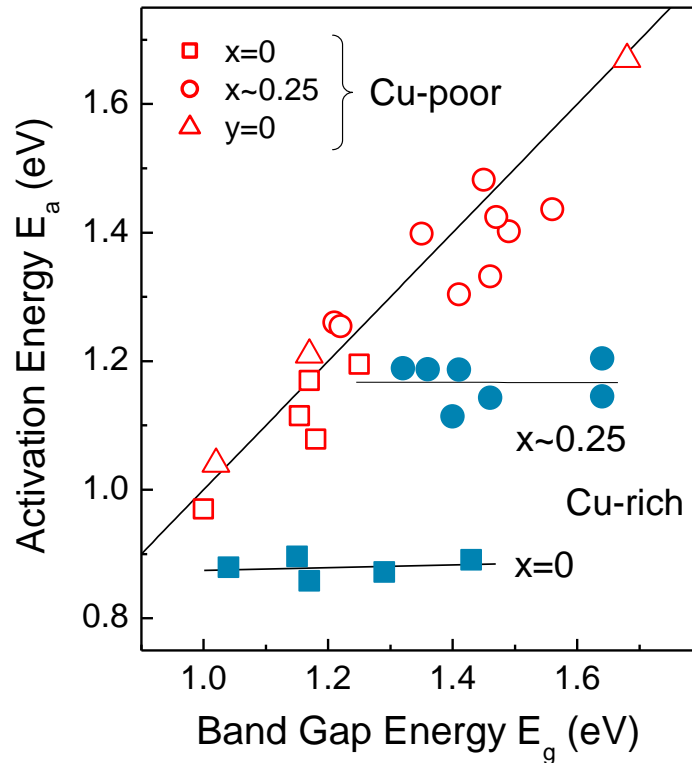


"Cu-rich": larger splitting of quasi-Fermi levels → higher V_{oc} potential

Finished cells: actual resulting V_{oc} drops at higher Cu-contents

⇒ splitting of quasi-Fermi levels not fully exploited for high Cu-contents

Recombination path in Cu-poor and "Cu-rich"



Cu-poor cells dominated by SCR recombination
"Cu-rich" by interface recombination

Turcu, Pakma, Rau, Appl. Phys. Lett., 80 (2002) 2598

A new process for "Cu-rich" solar cells

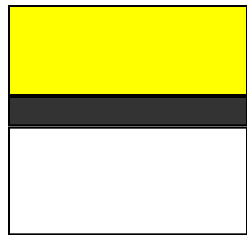
→ "Cu-rich" bulk with Cu-poor surface:

Cu, In, Se
co-deposition

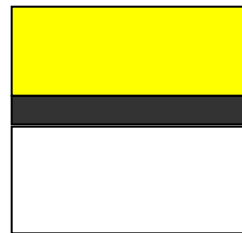
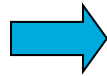
540 ° C

KCN
etching (10wt%, 5min)

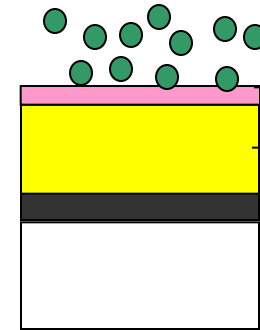
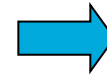
In-Se deposition
200 °C 1min + annealing



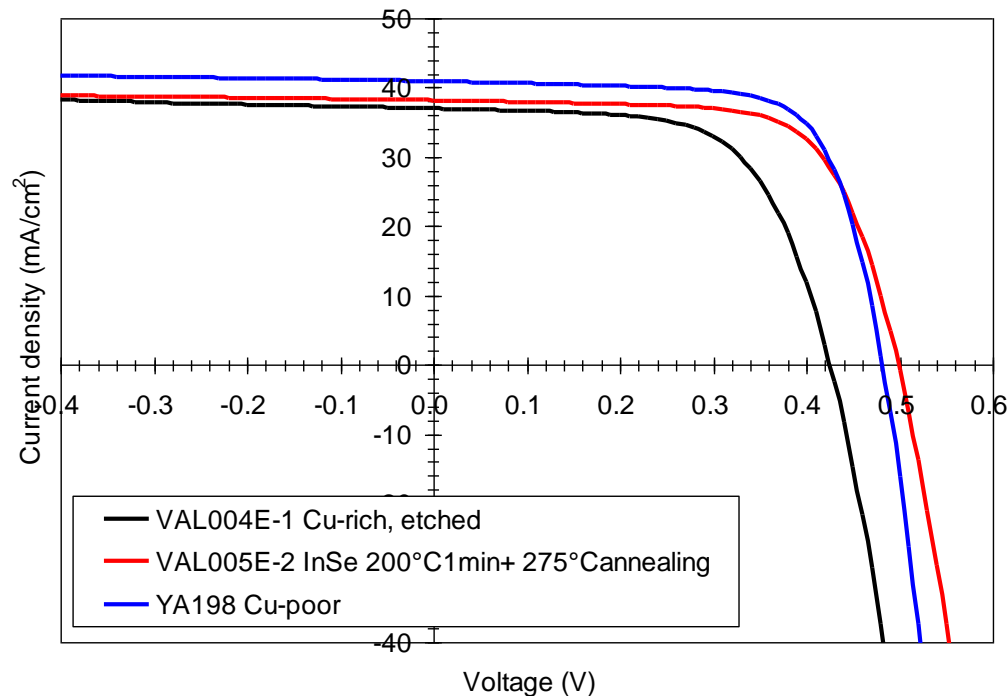
$\text{CuInSe}_2 + \text{Cu}_x\text{Se}$
Mo



Stoichiometric
 CuInSe_2
Mo



InSe 50nm
Stoichiometric
 CuInSe_2
Mo



currently best result:
13.1% efficiency
compare with
14.0% from Cu-poor

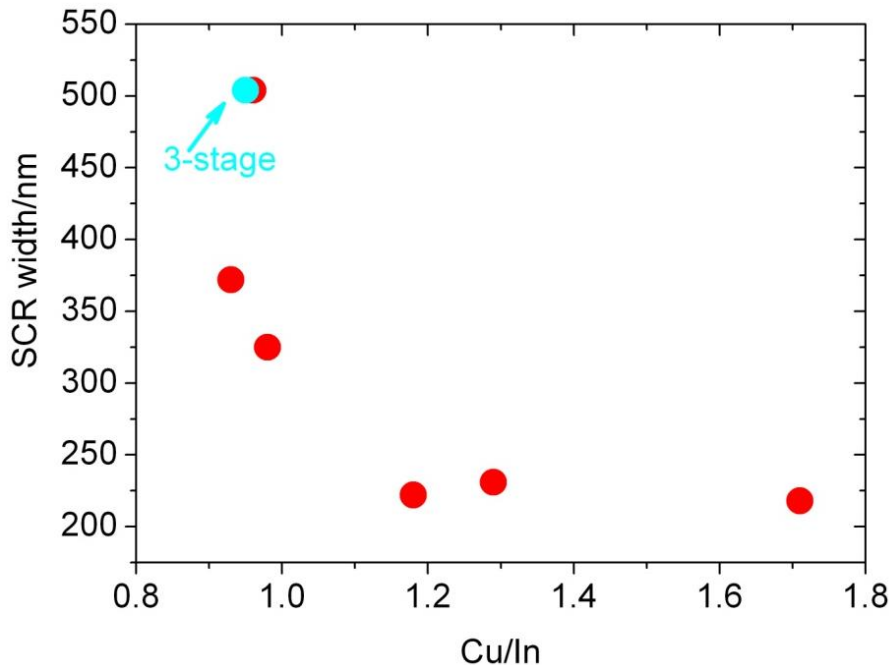
Depredurand, Aida, Siebentritt,
IEEE PVSC, Seattle, 337, 2011

Why is the current low?

one reason:

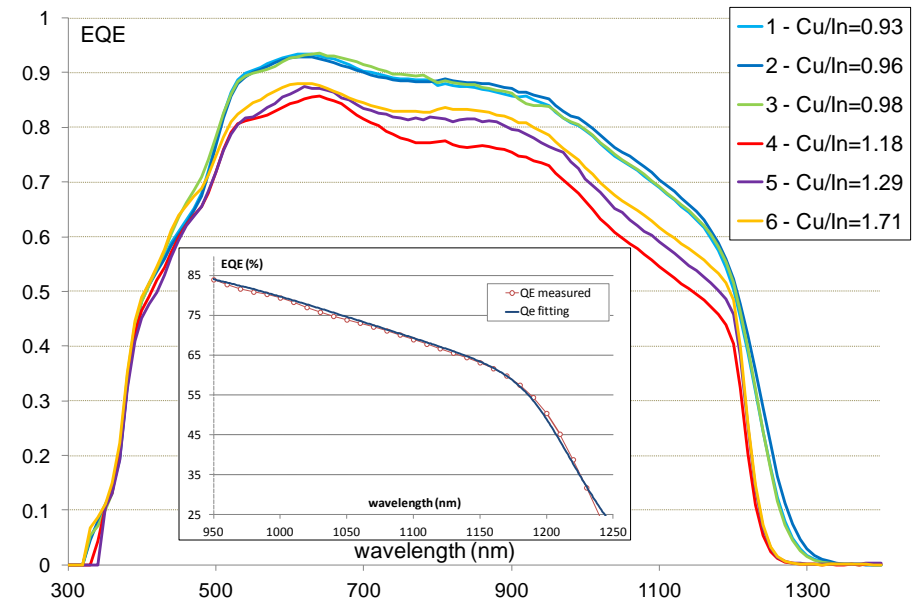
space charge width is too short

from capacitance measurements:



BUT: from QE \Rightarrow collection length

$$L_{eff} = W_{SCR} + L_{diff}$$



\Rightarrow space charge width is smaller

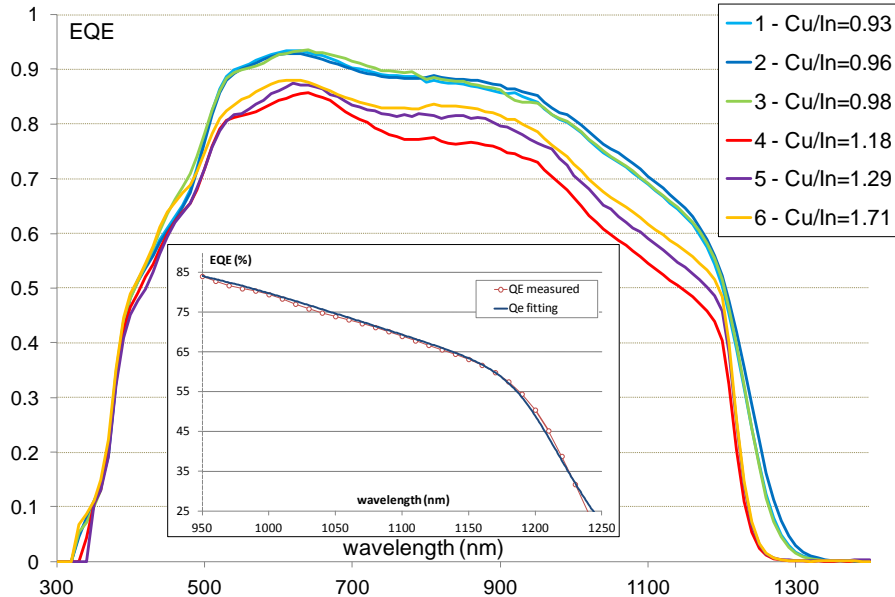
\Rightarrow the doping is too high!

$\Rightarrow L_{eff}(\text{Cu-rich}) \approx 3\mu\text{m}$

$L_{eff}(\text{Cu-poor}) \approx 2\mu\text{m}$

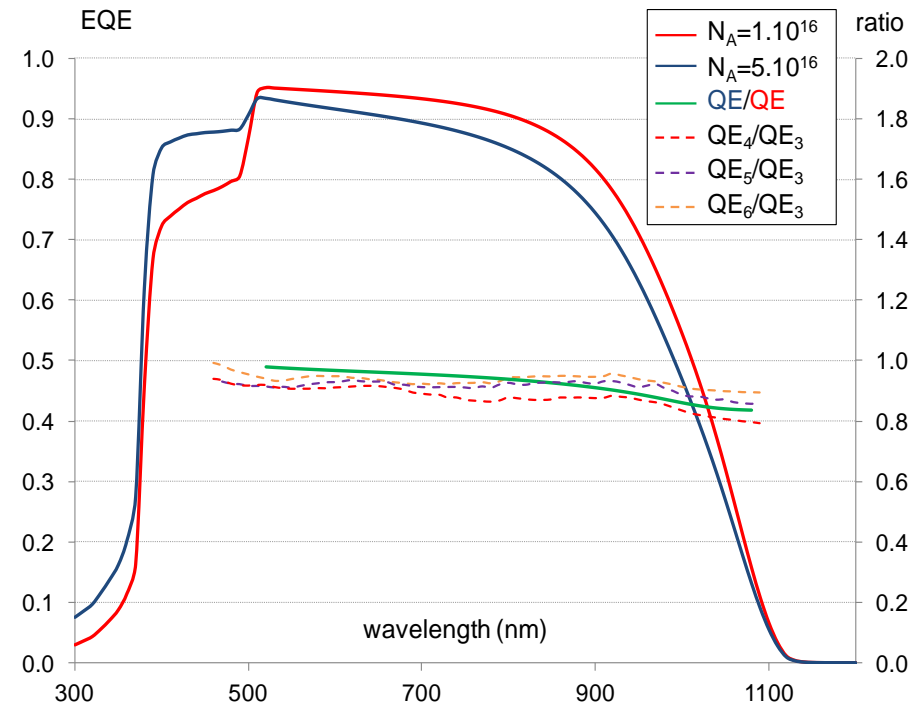
\Rightarrow collection length is better, in spite of shorter SCR width

Then why is the current lower?



⇒ QE lower by constant factor

SCAPS modelling:



⇒ reduced collection probability

⇐ high recombination in SCR
due to tunneling

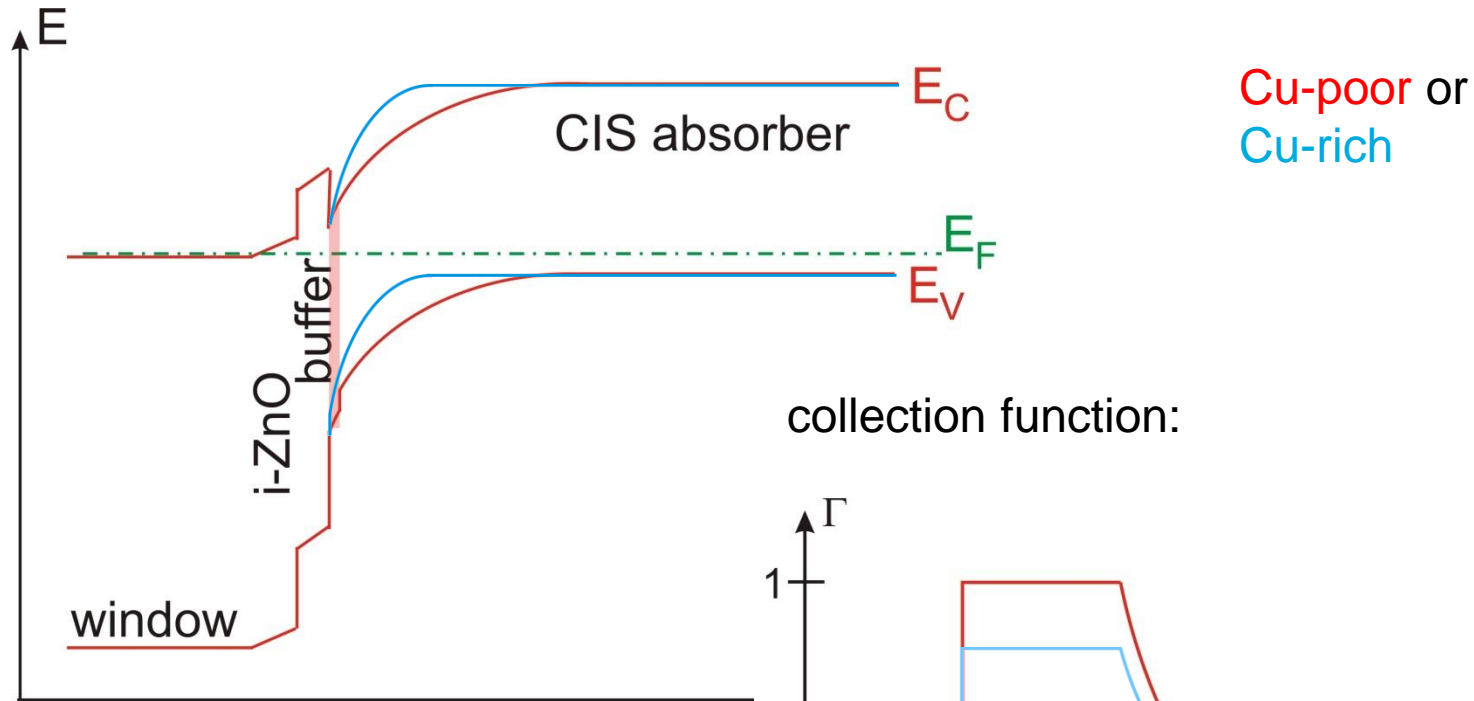


Depredurand, Tanaka, Siebentritt et al., submitted

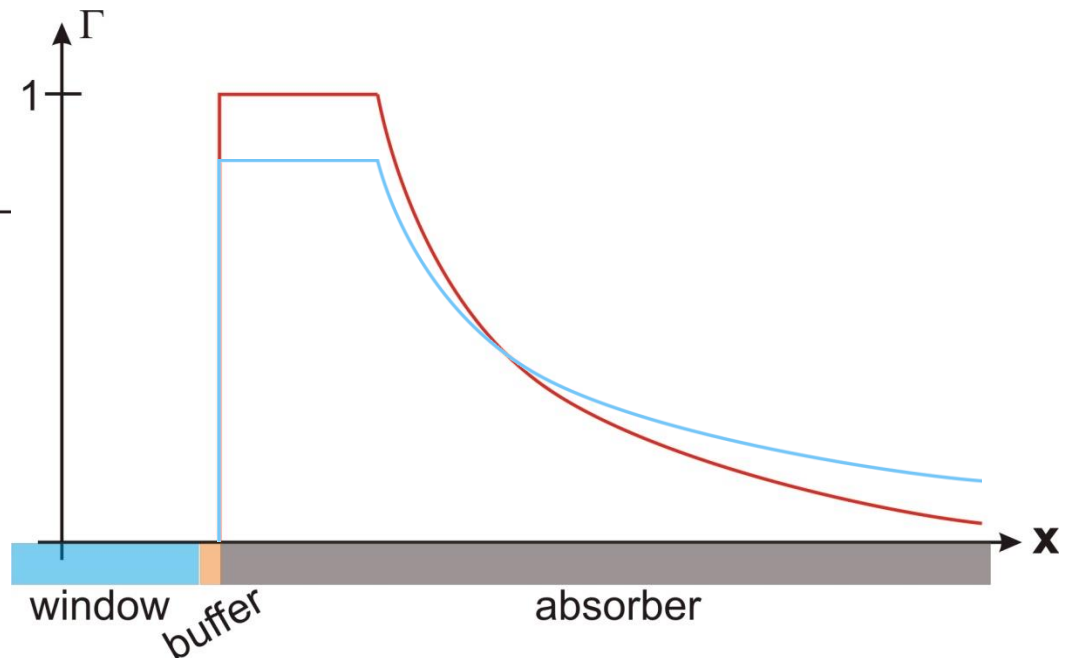
LUXEMBOURG

Cu-poor vs. Cu-rich

band diagram:



collection function:



Summary

- Indium and Gallium can be simultaneously electrodeposited with good compositional control over a large potential range.
- Laser annealing of nanocrystalline CuInSe_2 precursors for one second is sufficient to diffuse elements, grow grains, and develop semiconductor properties.
- Cu rich grown absorbers contain fewer defects and have a potential to produce higher V_{oc} devices than Cu poor grown absorbers.

Summary and Acknowledgements

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- Cu rich grown absorbers contain fewer defects and have a potential to produce higher V_{oc} devices than Cu poor grown absorbers.



Fonds National de la
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Thank you for your attention. Questions?



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